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In contrast to p/n homojunction development, CdTe heterojunction solar cells have been widely investigated since 1960, proceeding along two paths, according to CdTe conductivity type. For *n*-type CdTe single crystals and polycrystalline films, extensive work was carried out on heterojunctions with p-type Cu₂Te. In the early 1960s, n-type CdTe/p-type Cu₂Te devices having a structure analogous to the CdS/Cu₂S solar cell [10] were fabricated, by surface reaction of *n*-type single crystals or polycrystalline films in acidic aqueous solutions containing Cu salts for topotaxial conversion of CdTe to p-type Cu₂Te [11-15]. By the early 1970s, the best thin-film CdTe/Cu₂Te cells achieved cell efficiencies >7%, with $V_{\rm OC} = 550 \text{ mV}$, $J_{\rm SC} \sim 16 \text{ mA/cm}^2$ (60 mW/cm² irradiance), and FF = 50%, as reported by Justi *et al.* [16]. Interestingly, these cells utilized an underlying 5- μ m-thick *n*-type CdS layer to improve adhesion and electrical contact of the 20- μ mthick CdTe film on molybdenum substrates. Difficulty in controlling the Cu₂Te formation process, poor device stability in CdTe/Cu₂Te cells, and lack of a transparent *p*-type conductor ultimately shifted research emphasis to heterojunction structures employing *p*-type CdTe. Other work with *n*-type CdTe utilized Schottky barrier devices, formed by heating Pt or Au grids in contact with n-type CdTe single crystals [16] or electrodeposited CdTe thin films, with efficiencies approaching 9% [17].

For solar cells with single-crystal *p*-type CdTe, heterojunctions using stable oxides, such as In₂O₃:Sn (ITO), ZnO, SnO₂, and CdS have been more widely investigated. In these devices, the short wavelength spectral response is influenced primarily by the transmission of the heteropartner and low-resistance contact, collectively referred to as the window layer. Solar cells based on *p*-type CdTe single crystals with electron-beam-evaporated indium–tin oxide (ITO) window layers with efficiencies = 10.5% were developed by the Stanford group, 1977, with $V_{OC} = 810 \text{ mV}$, $J_{SC} = 20 \text{ mA/cm}^2$, and FF = 65% [18]. In 1987, cells made by the reactive deposition of indium oxide, In₂O₃, on *p*-type CdTe single crystals yielded total area efficiencies = 13.4%, with $V_{OC} = 892 \text{ mV}$, $J_{SC} = 20.1 \text{ mA/cm}^2$, and FF = 74.5% [19]. In this device, the CdTe crystal had a carrier concentration of $6 \times 10^{15}/\text{cm}^3$ and the CdTe (111) face was etched in bromine methanol prior to loading into vacuum for In₂O₃ deposition. The V_{OC} of this cell remains the highest ever reported for a CdTe device. Solar cells with ZnO window layers on *p*-type CdTe single crystals yielded poorer junction behavior, with efficiency <9% and $V_{OC} = 540 \text{ mV}$ [20].

Cells made by evaporating *n*-type CdS films onto single-crystal *p*-type CdTe were first prepared by Muller *et al.* in the mid-1960s [21, 22], yielding conversion efficiencies less than 5%. In 1977, Mitchell *et al.* reported a conversion efficiency = 7.9% with V_{OC} = 630 mV for a cell with 1-µm-thick CdS and an ITO transparent electrode [23]. The highest efficiency for a cell fabricated with thin-film CdS on *p*-type CdTe single crystal was reported by Yamaguchi *et al.* in 1977. Their cell utilized 0.5-µm-thick CdS deposited by chemical vapor deposition onto the (111) face of phosphorous-doped CdTe single crystals and gave 11.7% efficiency with V_{OC} = 670 mV [24].

Thin-film CdTe/CdS heterojunction solar cells have been fabricated in two different configurations, referred to as *substrate* and *superstrate*. In both configurations, light enters the cell through the transparent conducting oxide (TCO) and CdS films. However, in the superstrate cell, the TCO, CdS, and CdTe layers are sequentially deposited onto a glass superstrate, which also serves as the mechanical support for the cell, and light must pass through the supporting glass before reaching the CdS/CdTe junction. In the substrate configuration, the CdTe film is typically deposited first onto a suitable